

文章编号: 1006-9941 (2017)10-0810-07

Different Methods for Developing Relaxation Modulus Master Curves of AP-HTPB Solid Propellant

Walid M Adel, LIANG Guo-zhu

(School of Astronautics, Beijing University of Aeronautics and Astronautics, Beijing 100083, China)

Abstract: A comparative assessment of three different methods used for estimating and generating the relaxation modulus master curves of ammonium perchlorate-hydroxyl-terminated polybutadiene (AP-HTPB) solid propellant was carried out. These methods are the Williams-Landel-Ferry (WLF) method, the Arrhenius method, and the basic time-temperature superposition (TTS) method. The experimental data of the relaxation modulus for composite solid propellant were gathered by performing stress relaxation tests at constant strain level 10% for 1380 s at several temperature (-40, +20, +76 °C). The procedures and steps for evaluating the three methods was based on the same experimental data obtained for relaxation modulus to estimate the shift factors and then to generate the master curves of relaxation modulus and finally comparing the best fit method through the determination coefficient (R^2). The corresponding results indicate that the basic TTS method generated the best fit curve relative to the experimental data because this method is independent of external material constants and empirical equations in its application and it can be generally applicable to any viscoelastic material. However, in the applications of finite element software, the material constants are usually required to define the non-linear viscoelastic material model, in this case, the result of the present work demonstrates that both WLF and Arrhenius methods can produce satisfactory results, when appropriate constants are used, and the WLF method has proven to be more accurate and would be preferred in finite element analysis of AP-HTPB solid propellant.

Key words: ammonium perchlorate-hydroxyl-terminated polybutadiene (AP-HTPB); master curve; relaxation modulus; shift factor; Williams-Landel-Ferry (WLF) method; Arrhenius method; basic time-temperature superposition (TTS) method

CLC number: TJ55; O34; V512

Document code: A

DOI: 10.11943/j.issn.1006-9941.2017.10.004

1 Introduction

The solid propellant exhibits very complicated viscoelastic behavior and its mechanical properties (the relationship between tension and stress) and failure mechanisms are really sensible at the time of application loads, and temperature change. Relaxation modulus is one of the most important mechanical properties of viscoelastic materials which can be used to simulate the viscoelastic material behavior^[1]. The master curve of the relaxation modulus is defined as the relationship between the relaxation modulus and the reduced loading time and it is constructed based on a selected reference temperature of interest to which all data must be shifted using shift factors. The shift factor is defined as a unique parameter which takes into account both the temperature and strain rate effects of the temperature of interest relative to the reference temperature^[2]. Depending on a master curve, the mechanical properties and

the behavior of composite solid propellant can be predicted at different times and temperatures based on a limited set of experimental data. This anticipation is often performed through basic time-temperature superposition (TTS) principle.

The related publications in the extant literature, primarily concentrate on the application of TTS principle of solid propellant under different loading conditions. Chyuan analyzed numerically the structural integrity of HTPB solid propellant grains subjected to temperature loading^[3], ignition pressurization loading^[4], and Poisson's ratio varies under ignition pressure loading^[5], using a constitutive model based on the relaxation testing that employed the TTS and reduced integration by using WLF method. B. K. Bihari et al.^[6] used both method WLF and Arrhenius to determine the activation energy of composite solid propellant to predict the useful lifetime of solid propellant. Marimuthu et al.^[7] simulated the structural integrity of the solid propellant grains under gravity loading and inner pressure using relaxation tests and the TTS to specify the effective Poisson's ratio and Young's modulus. Moreover, Xu^[8] proposed a new method to get the relaxation modulus of HTPB propellants using the TTS in the temperature range of -50 °C to +35 °C. John Kim et al.^[1] estimated the master curve of the relaxation modulus for solid propellant according to the basic TTS method and WLF method, but without comparison be-

Received Date: 2017-04-24; **Revised Date:** 2017-05-30

Biography: Walid M Adel (1982-), PhD. Engaged in structural integrity of solid rocket motors. E-mail: walidadel1982@yahoo.com

Corresponding Author: LIANG Guo-zhu (1966-), professor. Engaged in teaching and research in the field of solid rocket motors. E-mail: lgz@buaa.edu.cn

tween the two methods. Some studies have been focused on comparing different TTS methods and methods to generate the relaxation modulus master curve, but using different materials for example Alex E. Alvarez et al.^[9] evaluated various methods and examples that can be satisfactorily applied to get the relaxation modulus master curves of hot-mix asphalt (HMA) by using three methods WLF, Arrhenius, and an optimization technique with the sum of square error (SSE) method. However, there is still limited information available about the difference between the methods that usually used to generate the relaxation modulus master curves of solid propellant, therefore, further researches should be performed to demonstrate the differences between these methods.

In this work, the specimen preparation and the stress relaxation tests at different temperatures are conducted. The three methods of generating the relaxation modulus master curves are discussed. Then the best fit model for the experimental data are presented.

2 Material and Experiment

2.1 Material and Specimens Preparation

The solid propellant used in this research is a heterogeneous propellant which consists of solid oxidizer particles ammonium perchlorate (AP) 67%, and metallic fuel particle aluminum powder (Al) 18%, dispersing in polymeric binder matrix (HTPB) forming the rest percentage. These gradients were mixed together, and then the propellant slurry was cast in special molds under vibration and vacuum with internal dimensions of 200 mm × 150 mm × 150 mm. Then the molds are placed in a large curing oven with temperature controlled at 60 °C, for a total curing time of 240 h. After curing the molds are cut into sheets with a uniform thickness. And so the test specimens (shown in Fig. 1) are produced using special cut press according to Joint Army-Navy-NASA-Air Force Propulsion Committee (JANNAF) standard^[10], and the dimensions of the test specimens are illustrated in Fig. 2. The actual thickness of the specimens is 11.04 mm less than the standard one



Fig. 1 HTPB solid propellant test specimen

due to some limitation of the sheet cutting machine. As a quality control step, the produced specimens are checked by a non-destructive method for voids like air bubbles or micro cracks by X-Ray to ensure the result of experimental data. After that, the accepted specimens were stored in desiccators at ambient temperature and relative humidity $RH \leq 30\%$. The mechanical and physical prosperities of the selected solid propellant (measured at room temperature) are listed in Table 1.

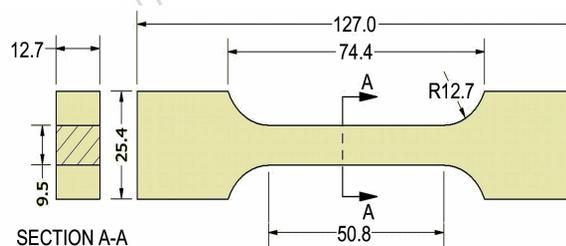


Fig. 2 Standard dimensions of the specimen (unit: mm)

Table 1 The mechanical and physical prosperities of the HTPB solid propellant

| Young's modulus /MPa | max. stress /MPa | max. strain /% | density / $\text{kg} \cdot \text{m}^{-3}$ | glass transition temperature / $^{\circ}\text{C}$ |
|----------------------|------------------|----------------|---|---|
| 3.40 | 0.724 | 34.5 | 1760 | -60 |

2.2 Stress Relaxation Test

To define the behavioral characteristics of a viscoelastic solid propellant accurately, its responses to an applied load or displacement must be determined as a function of strain rate, time, and temperature. These characteristics may be specified by means of creep or stress relaxation tests. The viscous nature of the mechanical behavior of a HTPB solid propellant is demonstrated by relaxation test which consists in subjecting a specimen to a constant elongation and measuring the evolution of the force $F(t)$ versus time, so the stress relaxation describes the time-dependent change in force due to applied displacement^[11]. The experiments were conducted using a computer controlled universal test machine (UTM) Zwick Z050 at different temperatures -40, +20 °C and +76 °C for 1380 s by maintaining constant strain level 10% during the whole time test and the initial tension rate was 10 mm · min⁻¹. Normally the researchers test the solid propellant specimens from -40 °C to +50 °C, but we select the high test temperature as +76 °C according to some engineering applications. Before the experimental tests the specimens are conditioned in an external environment chamber for three hours to ensure the thermal equilibrium, and also the relaxation tests were conducted in a digital control environmental temperature chamber with tolerance ± 0.1 °C of the set temperature point. The stress relaxation test shall also be repeated to check the result of the deflection su-

perposition during the stress relaxation phenomena^[12]. In order to ensure the consistency of the measured data, every experimental test was carried out on three specimens in the same conditions and the mean values of these results were taken as the final result. From the measured data, the stress relaxation modulus is calculated as indicated in Eq. (1).

$$E_R(t) = \frac{F(t)}{A\varepsilon} \quad (1)$$

where $E_R(t)$ is the time-dependent stress relaxation modulus, MPa; $F(t)$ is the measured time-dependent load, N; A is the cross section area of the specimen ($A=104.88 \text{ mm}^2$) and ε is the applied constant loading strain. The actual value of the applied strain is 0.098, a little less than 0.1, due to crosshead speed accuracy of the UTM.

Fig. 3 shows that the trend of the relaxation modulus decreases as time increases under various temperatures. The major change of the stress relaxation occurs in the first 120 s of the relaxation time after that the stress relaxation is still decreasing but with very slow rate. Also, it can be observed that the high effect of the low temperature and time on the stress relaxation curves for the composite solid propellant especially at the initial values, but this effect decreases as the relaxation time increases. For example, at $t=0 \text{ s}$ and at a constant strain level the difference percentage value between the relaxation modulus at -40°C and $+20^\circ\text{C}$ is 155.7%, while this value at $t=1380 \text{ s}$ is approximately 9.5%. These results totally represent that the current HTPB propellant is a viscoelastic material and exhibits the dependence of time-temperature behavior.

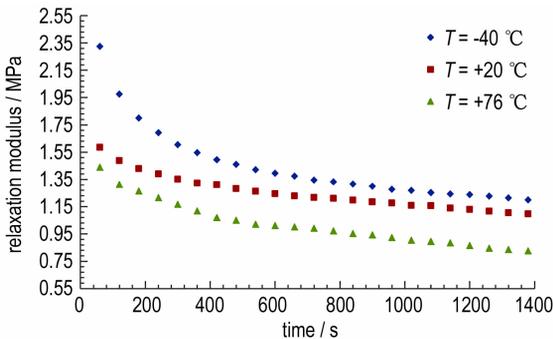


Fig. 3 Relaxation modulus at different temperatures and constant strain level $\varepsilon=10\%$

3 Methods for Generating the Master Curve

TTS principle has been used to get the master curves for several mechanical properties such as stress, strain, creep compliance and relaxation modulus against time or dynamic modulus against frequency^[13]. The various methods and models for producing the relaxation modulus master curves are discussed and evaluated in this section. These methods include the basic TTS method, the WLF method, and the Arrhenius method. The major difference between these methods is basi-

cally in the computation of the temperature shift factors. It should be emphasized here that the shift factor is the main driving force in the generation of any master-curve^[9].

3.1 Basic Time-Temperature Superposition Method

To generate the master curve of the relaxation modulus using the basic TTS method, it is necessary to plot the log relaxation modulus versus log time at different temperatures to obtain the shift factors of each time-temperature as shown in Fig. 4. The relationship between time and temperature can be written as the following equations^[11]:

$$E_R(t_0, T_0) = E_R(t_1, T_1) = E_R(t_2, T_2) \quad (2)$$

$$\log t_1 = \log t_0 + \log a_{T_1}^{T_0} \quad (3)$$

$$\log t_2 = \log t_0 + \log a_{T_2}^{T_0} \quad (4)$$

where $a_{T_1}^{T_0}$ and $a_{T_2}^{T_0}$ are the shift factors which can be obtained from the horizontal shift according to T_0 the reference temperature. In the present study, the reference temperature chosen is 20°C (293 K). Table 2 represents the shift factor obtained at various temperatures. Fig. 5 shows the TTS using the basic method.

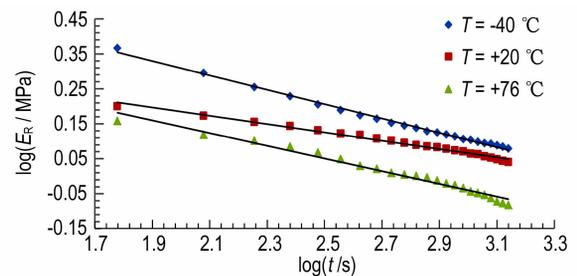


Fig. 4 $\log E_R$ vs. $\log t$

Table 2 Shift factors at different temperatures

| $T/^\circ\text{C}$ | $\log a_T$ |
|--------------------|------------|
| 76 | -0.354 |
| 20 | 0 |
| -40 | 0.477 |

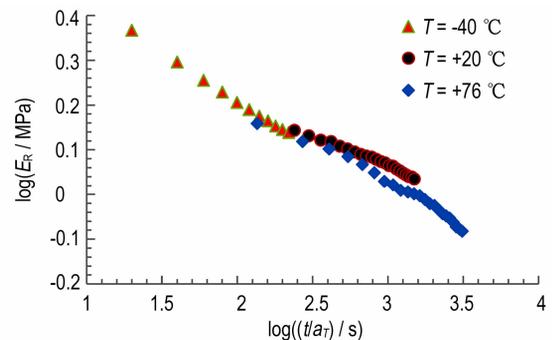


Fig. 5 Shifting process using the basic TTS method

A sole curve can be obtained from the above results called

the master curve, which gives the values of the relaxation modulus E_R against the values of the reduced time $\xi = t/a_T$ for various temperatures as shown in Fig. 6.

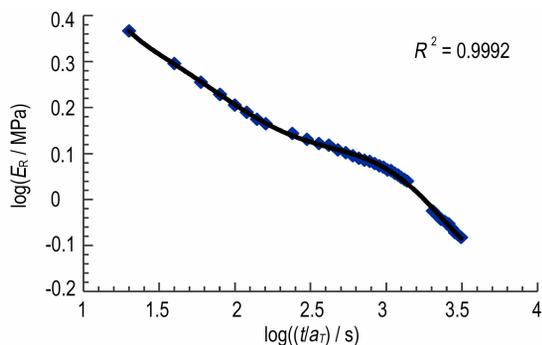


Fig. 6 Master curve using basic TTS method

3.2 The Williams-Landel-Ferry (WLF) Method

The WLF equation is an empirical equation that can be used to predict the behavior of the viscoelastic properties of the polymer at a wide range of temperatures. The WLF method is based on a free volume theory which is related to the macroscopic motion of the bulk material, and it is verified at temperatures above the glass transition temperature. The WLF TTS method for calculating the shift factor is shown in Eq. 5 [1, 2, 9]

$$\log a_T = \frac{-C_1(T-T_0)}{C_2+T-T_0} \quad (5)$$

where C_1 and C_2 are the material constants which are non-universal values, although they vary with the nature of the polymer system [14]. In order to calculate the shift factor at other temperatures, the material constants must be calculated by the following method. The linearized WLF equation can be rewritten as shown in Eq. 6 [1, 15].

$$\frac{1}{\log a_T} = \frac{1}{C_1} - \frac{C_2}{C_1(T-T_0)} \quad (6)$$

According to Eq. 6, if we plot $1/\log a_T$ as a function of $1/(T-T_0)$ we can extract C_2/C_1 from the resulting slope of the straight line and $1/C_1$ from the point of intersection with Y-axis.

Fig. 7 shows the linear relationship between $1/\log a_T$ and $1/(T-T_0)$, from Fig. 7 we can find $1/C_1 = -0.185$ and hence

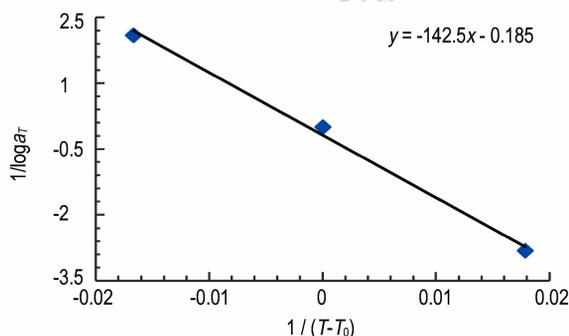


Fig. 7 Linearized WLF equation

$C_1 = -5.4$, and also we can find $C_2/C_1 = -142.5$, so $C_2 = 770.27$. Then the shift factor, according to WLF can be calculated at different temperatures according to the Eq. 4. Fig. 8 shows the TTS using WLF method, and Fig. 9 shows the master curve of the relaxation modulus using the same method.

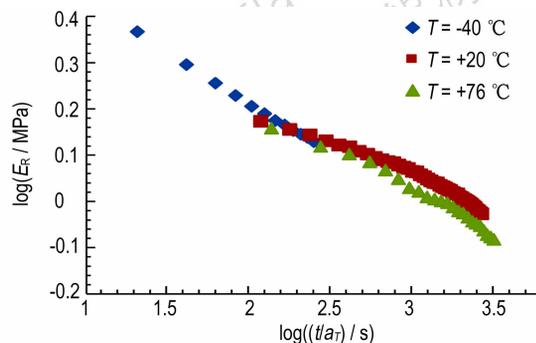


Fig. 8 Shifting process using WLF method

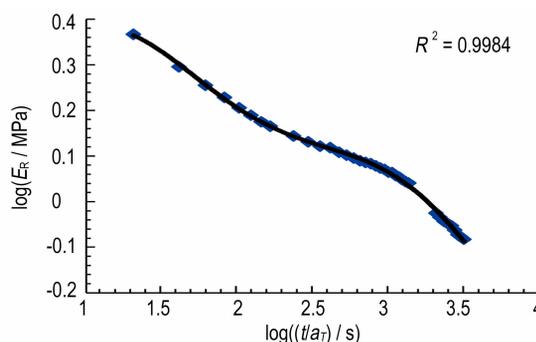


Fig. 9 Master curve using WLF method

3.3 The Arrhenius Method

The Arrhenius relationship is verified at temperatures under the glass transition temperature of the material, and the Arrhenius TTS method for shift factor calculation is shown in Eq. 7 [2, 9].

$$\log a_T = C_a \left(\frac{1}{T} - \frac{1}{T_0} \right) \quad (7)$$

where C_a is the material constant that is a function of the activation energy (E_a) and the universal gas constant (R), $C_a = E_a/2.303R$. T is the test temperature in K, and T_0 is the reference temperature (293 K). To obtain the material constant we must plot $\log a_T$ versus $(1/T - 1/T_0)$ and then the slope of the resulting line will be represent the material constant as shown in Fig. 10, so $C_a = 572.4$. Now, the shift factors at different temperatures can be obtained as shown in Fig. 11, and by using Eq. 6 the master curve of the relaxation modulus according to Arrhenius method can be plotted as shown in Fig. 12.

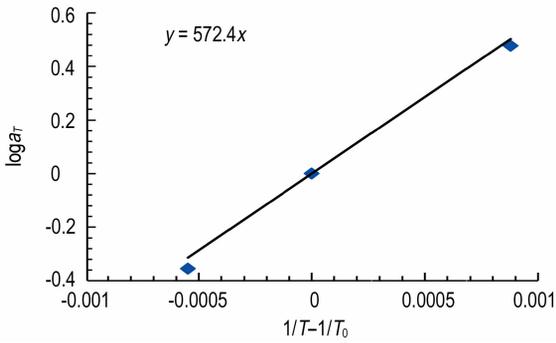


Fig. 10 Linearized Arrhenius equation

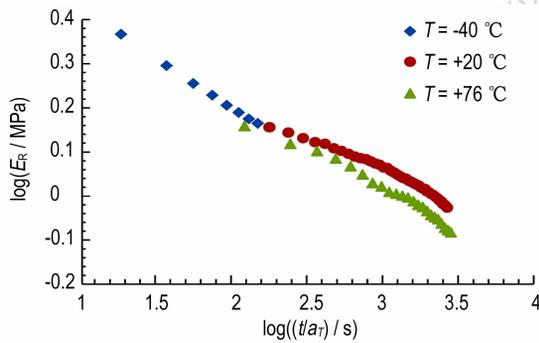


Fig. 11 Shifting process using Arrhenius method

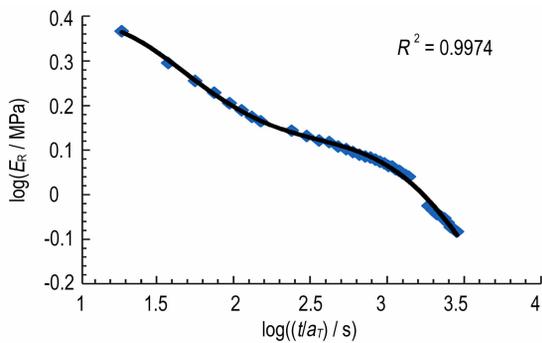


Fig. 12 Master curve using Arrhenius method

4 Results and Analysis

The master curves generated based on the basic TTS method, the WLF method, and the Arrhenius method using the same experimental relaxation test data presented in Fig. 3 are plotted in Fig. 13. The plots are for relaxation modulus as a function of reduced time in the log-log scale. All the curves are very similar, and this is an indication of the suitability of these methods for generating the relaxation modulus master curves of solid propellant. However, it should be stressed here that generation of satisfactory master curves and its accuracy is mostly dependent on the following factors:

Experimental test data consistency, which is a function of different variables, including both the parameters and the set-

up of the relaxation test, machine vibrations or noises, specimen homogeneity and uniformity, specimen fabrication for example the dimensions and parallelism of end surfaces of the specimen, and the variation of humidity conditions.

(2) The behavior of the shift factor variation with temperature.

(3) The condition of the polymeric material state.

(4) The heating rate applied to arrive to the desired temperature.

However, the analysis of the results demonstrated that the basic method for TTS has the highest accuracy of the relaxation modulus curve fitting given determination coefficient ($R^2 = 0.9992$), and this is not surprising because this method uses only the experimental data. Likewise, both the empirical methods can get acceptable results if appropriate material constants are used in these methods. Table 3 demonstrates the main statistical factors obtained during calculation of material constants, and then we can note the effect of these results on the accuracy of fitting curves. The sum of squared error (SSE), is a preliminary statistical calculation that leads to other data values. It is useful to be able to find how closely related those values are. The root mean squared error (RMSE) is the distance, on average, of a data point from the fitted line,

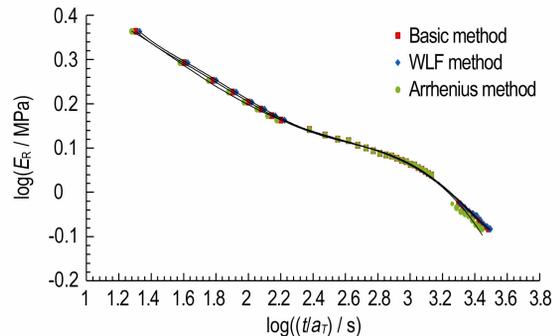


Fig. 13 Master curve of AP-HTPB solid propellant obtained using the three methods

Table 3 Statistics of analysis and material constant calculation for HTPB solid propellant

| parameter | WLF method | | Arrhenius method |
|-----------|--------------|----------------|------------------|
| | $C_1 = -5.4$ | $C_2 = 770.27$ | $C_a = 572.4$ |
| SSE | 0.0005247 | | 0.002355 |
| R^2 | 0.9989 | | 0.9932 |
| RMSE | 0.00156 | | 0.03431 |

Table 4 List of analysis and calculated parameters of master curves for HTPB solid propellant

| parameter | basic method | WLF method | Arrhenius method |
|-----------------------|--------------|------------|------------------|
| $a_{T=-40\text{ °C}}$ | 0.441906 | 0.431251 | 0.485892 |
| $a_{T=20\text{ °C}}$ | 1 | 1 | 1 |
| $a_{T=76\text{ °C}}$ | 3.002147 | 2.851846 | 3.184198 |
| R^2 | 0.9992 | 0.9984 | 0.9974 |

measuring along a vertical line. It seems that the data meet WLF method given R^2 coefficient ($R^2=0.9984$), while the data meet the Arrhenius method given R^2 coefficient ($R^2 = 0.9974$), less than the WLF one. Compared to the other two methods, the major advantage of the basic method is that it revolves around the actual experimental data without introducing any external constant when generation the master curve, while the WLF and Arrhenius methods use already existing empirical formulas which are dependent on externally determined material constants. Table 4 is a summary of the analysis and the calculated shift factors at different temperatures when generating the master curves of solid propellant. In Fig. 14, another type of master curve can be obtained from the previous calculation shows the relation between the shift factors corresponding to each temperature. It can be observed that the data meet WLF method given higher R^2 coefficient, and also given lower error than the data meet the Arrhenius method. In order to simulate the nonlinear viscoelastic behavior of the composite solid propellant under different loads and conditions using various commercial finite element codes (Ansys, Abaqus, Mark, etc.), one of the time-temperature dependent shift functions must be used, and the material constants such as C_1 , C_2 , and C_a must be entered as determined by the shift function specified and selected by the user in the code.

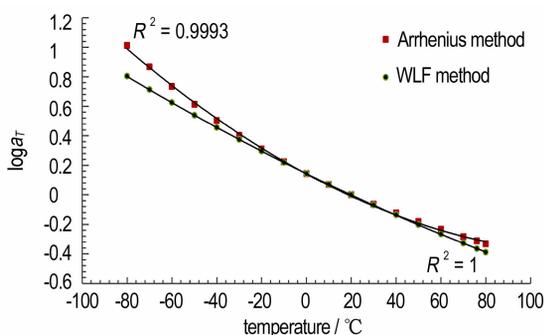


Fig. 14 Shift factors corresponding to each temperature

5 Conclusions

In this paper, a series of conventional relaxation tests have been performed using a universal test machine at a broad range of temperatures (-40 , $+20$, $+76$ °C) to evaluate and get the master curves of the relaxation modulus for AP-HTPB composite solid propellant as a purpose of reduced time and at a reference temperature of 20 °C. These master curves were generated according to three different methods (the basic, WLF, and Arrhenius) by using the TTS principle, and then comparative study was conducted to show the level of accuracy for each method. The following conclusions can be drawn:

(1) The basic method is the best method for generating a fit function for the relaxation modulus master curve followed by the WLF method and lastly, the Arrhenius method.

(2) The results presented here can be used as a reference for selecting the appropriate methods for generating the relaxation modulus master curve of AP-HTPB solid propellant.

(3) Most of the finite element software's need some material constants to define the time-temperature dependent material method, so the basic method cannot be used alone and some material constants must be calculated using empirical formulas like WLF or Arrhenius methods, and between these two methods, the WLF method would be recommended.

References:

- [1] Bohwi S, Jaehoon K. Estimation of master curves of relaxation modulus and tensile properties for solid propellant [J]. *Advanced Materials Research*, 2014, 871: 247–252.
- [2] Salvador N, Antonio M. New method for estimating shift factors in time-temperature superposition models [J]. *Journal of Thermal Analysis and Calorimetry*, 2013, 113: 453–460.
- [3] Chyuan S. Nonlinear thermoviscoelastic analysis of solid propellant grains subjected to temperature loading [J]. *Finite Elements in Analysis and Design*, 2002, 38(7): 613–630.
- [4] Chyuan S. Dynamic analysis of solid propellant grains subjected to ignition pressurization loading [J]. *Journal of Sound and Vibration*, 2003, 268(3): 465–483.
- [5] Chyuan S. Studies of Poisson's ratio variation for solid propellant grains under ignition pressure loading [J]. *International Journal of Pressure Vessels and Piping*, 2003, 80(12): 871–877.
- [6] Bihari B K, Waniet V S, Rao N P N, et al. Determination of activation energy of relaxation events in composite solid propellants by dynamic mechanical analysis [J]. *Defense Science Journal*, 2014, 64(2): 173–178.
- [7] Marimuthu R, Nageswara B. Development of efficient finite elements for structural integrity analysis of solid rocket motor propellant grains [J]. *International Journal of Pressure Vessels and Piping*, 2013, 111: 131–145.
- [8] Xu J, Ju Y, Han B. et al. Research on relaxation modulus of viscoelastic materials under unsteady temperature states based on TTSP [J]. *Mechanics of Time-Dependent Materials*, 2013, 17(4): 543–556.
- [9] Lubinda F, Alex E, Geoffrey S. Evaluating and comparing different methods and models for generating relaxation modulus master-curves for asphalt mixes [J]. *Construction and Building Materials*, 2011, 25(5): 2619–2626.
- [10] Davenas A. Solid rocket propulsion Technology [M]. First English Edition, New York, Pergamon Press, 1993, Chapter 6.
- [11] Ferry J D. Viscoelastic properties of polymers [M]. 3rd Edition, John Wiley and Sons, 1980.
- [12] Zheng Z, Zhang R. Implementation of viscoelastic material model to simulate relaxation in glass transition [C] // Excerpt from the Proceedings of the 2014 COMSOL Conference in Boston, 2014.
- [13] Carlrton J M. SRM propellant and polymer materials structural test program. NASA Technical Paper 2821, 1988.

- [14] KGNC A, Burgoyne C J. Time-temperature superposition to determine the stress-rupture of aramid fibres [J]. *Applied Composite Materials*, July 2006, 13(4) : 249-264.
- [15] ROBERT F L. Viscoelastic properties of rubberlike composite propellants and filled elastomers [J]. *ARS Journal*, 2015, 31(5) : 599-608.

构建 AP-HTPB 固体推进剂松弛模量主曲线的不同方法

Walid M Adel, 梁国柱

(北京航空航天大学宇航学院, 北京 100083)

摘要: 分析了建立高氯酸铵-端羟基聚丁二烯 (AP-HTPB) 固体推进剂松弛模量主曲线的三种不同方法 (Williams-Landel-Ferry (WLF) 方法、Arrhenius 方法和基本的时间-温度叠加 (TTS) 方法)。通过不同温度 (-40, +20, +76 °C) 和 10% 常应变下 1380 s 的应力松弛试验得到了复合固体推进剂松弛模量。通过评估松弛模量, 给出转移因子和松弛模量主曲线, 最后通过确定系数 (R^2) 得出了最佳的拟合方法。结果表明, 基本 TTS 方法可给出最佳拟合曲线, 因为该方法不依赖外部材料常数和应用中的经验方程, 而且该方法可以适用于任何黏弹性材料。但是, 在应用有限元软件时, 经常要求用材料常数来定义非线性黏弹性材料模型, 这种情况下, 目前研究表明, 当给出合适的常数后, WLF 和 Arrhenius 两种方法均可给出满意的结果, 而且 WLF 方法更为准确, 因此在 AP-HTPB 固体推进剂有限元分析中倾向于采用这种方法。

关键词: 高氯酸铵-端羟基聚丁二烯 (AP-HTPB); 主曲线; 松弛模量; 转移因子; Williams-Landel-Ferry (WLF) 方法; Arrhenius 方法; 时间-温度叠加 (TTS) 方法

中图分类号: TJ55; O34; V512

文献标志码: A

DOI: 10.11943/j.issn.1006-9941.2017.10.004